

Dynamics of Lithium Dendrite Growth and Inhibition - Pulse Charging Experiments and Monte Carlo Calculations

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Abstract: Short-circuiting via dendrites compromises the reliability of Li-metal batteries. Since dendrites ensue from instabilities inherent to electrodeposition they should be amenable to dynamic control. Here we report experiments in a scaled coin cell prototype where the average lengths of dendrite populations were shortened ~ 2.5 times upon charging with 1 ms rectangular pulses followed by 3 ms rest periods. Monte Carlo simulations dealing with Li^+ diffusion and electromigration show that experiments involving 20 ms pulses, in contrast, were ineffective because prolonged migration in the strong electric fields converging to high-curvature dendrite tips generates depleted layers that cannot be replenished by diffusion after 60 ms rest periods. Since the application of pulses shorter than the characteristic time for capacitively polarizing electric double layers in our system ($\tau_c \sim 3$ ms) would approach DC charging, we infer that dendrite propagation may be inhibited (albeit not suppressed) by pulsing at appropriate frequency ranges.

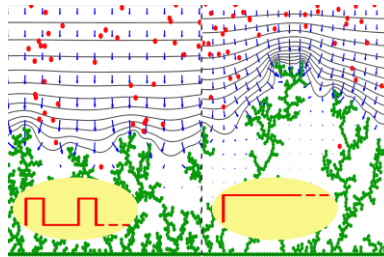


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The specific high energy and power capacities of lithium-metal (Li^0) batteries are ideally suited to power portable devices, and as storage units for intermittent renewable energy sources.¹⁻⁷ Li^0 , the lightest and most electropositive metal, should be the optimal anode material for rechargeable batteries if it were not for the fact that such devices fail unexpectedly by short-circuiting via the dendrites that grow across electrodes upon recharging.⁸⁻⁹ This phenomenon triggers a series of adverse events that start with overheating, followed by the thermal decomposition and ultimately the ignition of the organic solvents used in such devices. This flaw represents a major safety issue.¹⁰⁻¹²

Li^0 dendrites have been imaged, probed and monitored with a wide array of techniques.^{4-5,}¹³ Their formation was analyzed¹⁴⁻¹⁵ and simulated at various levels of realism.^{9, 16-17} Numerous empirical and semi-empirical strategies were employed for mitigating the formation of Li^0 dendrites, mostly based on variations of electrode materials and morphologies, and operational conditions.² Thus, there are reports on the effect of current density,¹⁸⁻²⁰ electrode surface morphology,¹⁰ solvent and electrolyte composition,²¹⁻²⁴ electrolyte concentration,¹⁸ evolution time,²⁵ the use of powder electrodes,²⁶ and adhesive lamellar block copolymer barriers²⁷ on dendrite growth. Conceivably, further progress in this field could accrue from the deeper insights on the mechanism of dendrite propagation gained by increasingly realistic, and properly designed experiments and modeling calculations.^{23, 28} We considered that Li^0 dendrite nucleation and propagation are intrinsic to electrodeposition as a dynamic process under non-equilibrium conditions.^{5, 14} Also, that in contrast with purely diffusive crystal growth, Li-ion (Li^+) electromigration is an essential feature of electrolytic dendrite growth.²⁹ More specifically, we envisioned that runaway dendrite propagation could be slowed by the relaxation of the steep Li^+ concentration gradients and strong electric field that develop around dendrite tips during charging. This is not a new strategy,³⁰ but to our knowledge the quantitative *statistical* impact of pulses of variable duration on dendrite length has not been reported before. Herein we report experiments on dendrite growth in a scaled coin cell prototype of novel design fitted with Li^0 electrodes under rectangular cathodic pulses of variable rest periods and frequencies in the kHz range. Our cell mimics the geometry and aspect ratio of commercial coin cells but it is

scaled-up to allow for the visual observation of dendrites. The effects of pulsing on stochastic events such as dendrite nucleation and growth are quantified for the first time on the basis of statistical averages over experimental dendrite length distributions. We also present novel coarse-grained Monte Carlo model calculations which, by dealing explicitly with Li^+ migration in time-dependent non-uniform electric fields, provide valuable insights into the underlying phenomena. Our findings are expected to guide the implementation of safer charging protocols for commercial batteries.

Our experiments were performed in a manually-fabricated electrolytic cell that provides for *in situ* observation of the dendrites grown on the perimeter of the electrodes at any stage (Fig. 1). The cell consists of two Li^0 foil disc electrodes (1.59 cm diameter) separated 0.32 cm by a transparent acrylic ring. The cell was filled with 0.4 cm^3 of 1 M LiClO_4 in propylene carbonate (PC) as electrolyte. All operations were conducted in an argon-filled (H_2O , $\text{O}_2 < 0.5$ ppm) glove-box. Arrays of multiple such cells were simultaneously electrolyzed with trains of 2 mA cm^{-2} pulses of variable duration: t_{ON} , and idle ratios: $\gamma = t_{\text{OFF}}/t_{\text{ON}}$, generated by a programmable multichannel charger. After the passage of 48 mAh (173 Coulombs) through the cells, the lengths of 45 equidistant dendrites around the cell perimeter were measured through the acrylic separator using a Leica M205FA optical microscope. Since in our cell dendrites propagate unimpeded, i.e. in the absence of a porous separator, our experiments may provide more adverse conditions for controlling dendrite propagation than in actual commercial cells. Further details can be found in Experimental Details, Supporting Information, SI.

The lengths and multiplicities $[\lambda_i, p_i]$ of the 45 dendrites measured in series of experiments performed at $t_{\text{ON}} = 1$ and 20 ms, $\gamma = 0$ (DC), 1, 2 and 3, have been divided in 8 bins and are shown as histograms in Fig. S1 (SI). Dendrite lengths typically spanned the 200 μm – 3000 μm range. Their average length α , defined by equation (E 1):

$$\alpha = \frac{\int p_i \lambda_i}{\int p_i} \quad (\text{E } 1)$$

represents a figure of merit appropriate (vs. the length of a single dendrite chosen arbitrarily) for appraising the effect of pulsing on the outcome of a stochastic processes such as the inception and propagation of dendrites. The resulting α values, normalized to the largest α in

each set of experiments, are shown as blue bars as functions of γ for $t_{ON} = 1$ and 20 ms pulses in Fig. 2. It is immediately apparent that the application of $[t_{ON} = 1 \text{ ms}; t_{OFF} = 3 \text{ ms}]$ pulse trains reduces average dendrite lengths by ~ 2.4 times, whereas $t_{ON} = 20 \text{ ms}$ pulses are rather ineffective at any γ .

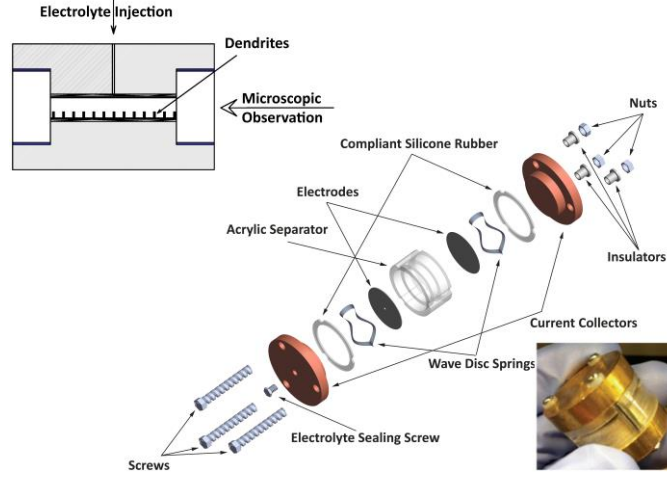


Fig. 1 – Top down: cross section view, exploded view and physical appearance of the cell.

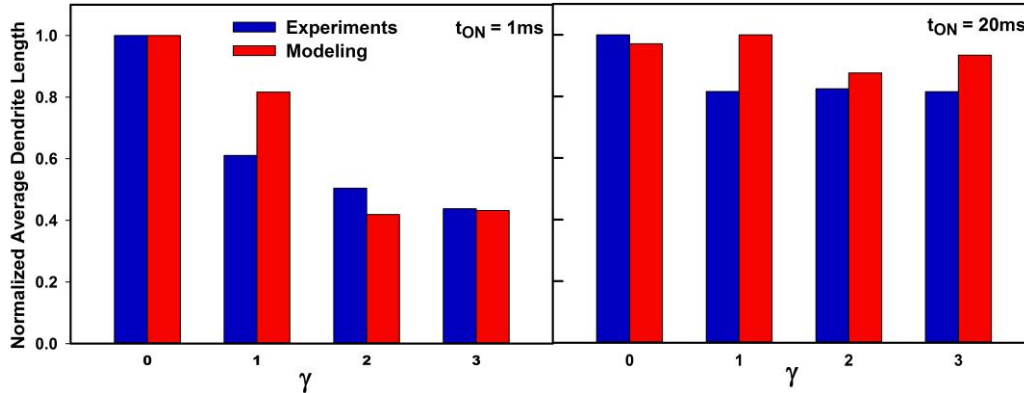


Fig. 2 - Pulse charging effects on the average dendrite length α over a population of 45 dendrites grown on the perimeter of the Li^0 cathode. $\gamma = t_{OFF}/t_{ON}$ is the idle ratio.

Basic arguments clarify the physical meaning of the $t_{ON} \sim 1 \text{ ms}$ time scale. The mean diffusive (MSD) displacement of Li^+ ions: $\text{MSD} = (2 D_+ t)^{1/2}$, (D_+ is the experimental diffusion coefficient of Li^+ in PC) defines the thickness of the depletion layers created in the vicinity of the

cathode surface via Faradaic reduction of Li^+ that could be half-replenished via diffusion during t rest periods.¹⁵ Notice that MSD is a function of time and only depends on a property of the system (D_+), i.e. it is independent of operating conditions, such as current density. From the Einstein relationship: $D_+ = \mu_+ (RT/F)^{31}$ (μ_+ = is the mobility of Li^+ in PC), the electric fields $|E|_c$ at which Li^+ electromigration displacements: $\text{EMD} = \mu_+ |E|_c t$, match MSD are given by equation (E 2):

$$|E|_c = (2 RT/F)^{1/2} \mu_+^{-1/2} t^{-1/2} \quad (\text{E } 2)$$

Thus, with $(2 RT/F) = 50 \text{ mV}$ at 300 K, $\mu_+ = 1 \times 10^{-4} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$, $t = 1 \text{ ms}$, we obtain $|E|_c = 707 \text{ V cm}^{-1}$, which is considerably stronger than the homogeneous field between (flat) electrodes at the beginning of experiments: $|E|_0 \sim V_0/L = 9.4 \text{ V cm}^{-1}$. However, cathode flatness and field homogeneity are destroyed upon the inception of dendrites, in whose sharp (i.e. large radii of curvature) tips local fields arise that are much stronger than $|E|_0$.^{15, 32} Under such conditions, Li^+ will preferentially migrate to the tips of advancing dendrites rather than to flat or concave sectors of the cathode surface.^{14-15, 33-35} Furthermore, given the stochastic nature of dendrite propagation, one should expect a distribution of tip curvatures. Hence, the mean field condition: $\text{EMD} \leq \text{MSD}$, at specified t_{ON} values may be realized by a subset of the population of dendrites. On sharper dendrites the inequality $\text{EMD} > \text{MSD}$ will apply at the end of t_{ON} pulses. Thus, larger $|E|_c$ values extend the $\text{EMD} \leq \text{MSD}$ conditions to dendrites possessing sharper tips, i.e. to a larger set of dendrites that could be controlled by pulsing. Note, however, the weak $|E|_c \propto \mu_+^{-1/2} \propto \eta^{-1/2}$ dependence on solvent viscosity η .

From this perspective, since $|E|_c \propto t^{-1/2}$, the application of longer charging pulses will increase the width of the depletion layers over a larger subset of dendrites to such an extent that such layers could not be replenished during rest periods, thereby magnifying the conditions favoring dendrite propagation. The preceding analysis suggests that shorter t_{ON} periods should be increasingly beneficial. Could t_{ON} be shortened indefinitely? The answer is no, because at sufficiently high pulsing frequencies charging will approach a DC regime. The transition from pulsed to DE charging takes place when t_{ON} becomes shorter than the characteristic times of the transients associated with the capacitive polarization of electrochemical double layers upon

application of a step voltage. A fundamental analysis of such transients³⁶⁻⁴⁰ revealed that the primary time scale for diffuse-charge dynamics is given by τ_c , equation (E 3):

$$\tau_c = \lambda_D L / D_+ \quad (\text{E } 3)$$

where $\lambda_D = (\epsilon k_B T / 2 z^2 e^2 C_0)^{1/2}$ is the Debye screening length, L the inter-electrode gap and D_+ the Li^+ diffusion coefficient. In our system, for $C_0 = 1 \text{ M Li}^+$ solutions in PC ($\epsilon = 65$), $D_+ = 2.58 \times 10^{-6} \text{ cm}^2 \text{ s}^{-1}$, at 298 K, we get: $\lambda_D = 0.27 \text{ nm}$ and $\tau_c = 3.3 \text{ ms}$. In physical terms, $\tau_c = 3.3 \text{ ms}$ is the characteristic time for charging the capacitors associated with diffuse double layers. Capacitive charging implies the partial segregation of anions from cations, a phenomenon that increases the electric field close to the cathode surface and, therefore, triggers Faradaic deposition. Under t_{ON} pulses increasingly shorter than τ_c , most of the initial current will be capacitive, thereby implying that Faradaic currents will continue to circulate during the rest periods. In other words, the decreasing amplitude of polarization oscillations under trains of t_{ON} pulses shorter than τ_c will gradually converge to steady DC.

In summary, shorter t_{ON} pulses are increasingly beneficial in inhibiting dendrite propagation, but they are loosely bound by the condition $t_{ON} \geq \tau_c$. The underlying reason is that shorter t_{ON} pulses will inhibit dendrite at earlier propagation stages, where the curvature of most dendrite tips has not reached the magnitude at which the local electric fields lead to the $\text{EMD} > \text{MSD}$ runaway condition. Notice that the stage at which dendrite propagation can be controlled by pulsing relates to the curvature of tip dendrites, which is a morphological condition and therefore independent of current density. Higher current densities, however, will shorten the induction period preceding dendrite nucleation.³⁴

These ideas were cast and tested in a coarse-grain Monte Carlo model that, in accord with the preceding arguments, deals explicitly with ion diffusion, electromigration and deposition. It should be emphasized that our model is more realistic than those previously reported⁹ because it takes into account the important fact that dendritic growth should be critically dependent on the strong electric fields that develop about the dendrites tips upon charging.⁴¹ In fact, the key role of electromigration in dendrite propagation has been recently demonstrated by the smooth Li^0 cathode surfaces produced in the presence of low concentrations of non-reducible cations, such as Cs^+ that, by preferentially accumulating on dendrite tips, neutralize the local

electric fields and deflect Li^+ toward the flat cathode regions.³ Thus, both the population of electroactive Li^+ species and the electric vector field within the partially depleted, polarized double layers surrounding the cathode is established from the competition of ion diffusion with electromigration rather than with Li^+ deposition. Given the typically small overpotentials for metal ion reduction on metallic electrodes,³¹ we consider that the effect of the applied external voltage on dendrite growth operates via the enhancement of Li^+ migration rather than Li^+ reduction. In our model dendrite nucleation is a purely statistical phenomenon, i.e., nucleation occurs because there is a finite probability that two or more Li^+ ions are successively reduced at a given spot on the cathode surface. Once a dendrite appears, a powerful positive feedback mechanism sets in. The enhanced electric field at the tip of the sharp dendrite attracts Li^+ ions faster, thereby accelerating dendrite propagation and depleting the solution of Li^+ in its vicinity. Thus, the anisotropic concentration gradients observed nearby growing dendrites may be a consequence of the statistical onset of dendrites. In other words, we consider that simultaneity does not imply causality, and that the fast depletion of Li^+ upon dendrite inception is an effect rather than a cause of dendrite nucleation. We wish to emphasize, however, that we are aware our arguments and modeling effort do not exclude alternative proposals to account for dendrite nucleation. Furthermore, we also want to underscore the fact that the validity of one view or another on the mechanism of dendrite nucleation may not affect our interpretation that the effects of pulsing on dendrite propagation arise from the competition between ion diffusion and electromigration once dendrites appear. Due to the very high computational cost of atomistic modeling, we simulate the processes taking place in a domain that is smaller than the actual cell. Its actual dimensions approximately correspond to the depth of a depletion boundary layer at the cathode. Calculations are based on the scaled parameters shown in Table 1. Scaling was guided by the following considerations. Because our calculations aim at reproducing the frequency response of our experiments, simulation time was set to real time. The length scale, however, is dictated by computational cost. Thus, if diffusional displacements in real time would not exceed the size of our domain ($L = 16.7 \text{ nm}$, Table 1) the diffusion coefficient should be scaled accordingly. The adopted $D_+ = 1.40 \times 10^{-10} \text{ cm}^2/\text{s}$ value in fact leads to $\text{MSD} \sim 0.3 L$ after 1 ms. A voltage difference of $(V_{\text{top}} - V_{\text{cathode}}) = 14.7 \text{ mV}$ between the

electrodes separated by $L = 16.7$ nm generates electric fields in which electromigration is competitive with diffusion. The implemented two-dimensional Monte Carlo algorithm calculates the trajectories of individual Li^+ ions via random diffusion and electro-migration under time and position-dependent electric fields $\vec{E}(x, y, t)$.

Table 1 –Parameters used in the Monte Carlo calculations

Domain size L	16.7 nm \times 16.7 nm
Δt (integration step)	1 μs
V_{cathode}	0 V
V_{top}	47.4 mV
D_+ (Li^+ diffusion coefficient)	$1.40 \times 10^{-10} \text{ cm}^2/\text{s}$
μ_+ (Li^+ mobility)	$5.54 \times 10^{-9} \text{ cm}^2/\text{Vs}$
Li^+ radius	1.19 Å
Free Li^+ ions	50
Maximum Li^0 atoms	600

By assuming that Li^+ ions reach stationary velocities instantaneously, their mean displacements are given by:

$$\vec{r}_i(t + \Delta t) - \vec{r}_i(t) = \sqrt{2 D_+ \Delta t} \vec{g} + \mu_+ \vec{E} \Delta t \quad (\text{E } 4)$$

The first and second terms in the RHS of E3 are the mean displacements due to ionic diffusion and electromigration, respectively. \vec{g} is a normalized 2D vector representing random motion via diffusion, Δt is the computational time interval, and \vec{E} is the electric field vector. By normalizing displacements relative to the inter-electrode separation, L , (E 4) transforms into (E 5):

$$\vec{\xi}(t + \Delta t) = \vec{\xi}(t) + \theta \vec{g} + \vec{\eta} \quad (\text{E } 5)$$

Dendrite lengths λ_i were evaluated as their height $\alpha_i(t)$ above the surface of the electrode:

$$\lambda_i(t) = \max_{k=1:n} \vec{\xi}_k(t) \cdot \vec{j} \quad (\text{E } 6)$$

\vec{j} is the unit vector normal to the surface of the electrode and n is the total number of lithium atoms incorporated into the dendrite.

By using the Einstein relationship above, the equation of motion becomes:

$$\vec{r}(t + \Delta t) - \vec{r}(t) = \sqrt{2 D_+ \Delta t} \vec{g} + \frac{F}{RT} D_+ \Delta t \vec{E} \quad (\text{E } 7)$$

a function of $D_+ \Delta t$.

By neglecting electrostatic ion-ion interactions, given that they are effectively screened since $\lambda_D = 0.27$ nm is smaller than the average interionic separation $R_{i,j} = 1.2$ nm, $\overrightarrow{E(x,y,t)}$ is computed using Laplace's equation:

$$\frac{\partial^2 \phi}{\partial x^2} + \frac{\partial^2 \phi}{\partial y^2} = 0 \quad (\text{E } 8)$$

Thus, the electric field is determined by the evolving geometry of the equipotential dendritic cathode. In reality, the concentration gradients that develop in the depleted boundary layers would lead to even greater electric field enhancements than reported herein. The inclusion of ion-ion interactions and charge imbalances in Monte Carlo calculations would be too costly computationally. We consider, however, that the inclusion of a variable electric field represents a significant advance over previous models.⁹

Calculated dendrite heights were quantified by dividing the x-axis (parallel to the surface of the cathode) in four sectors. Here, 'dendrite height' in each sector is the height of the Li^0 atoms furthest from the electrode. To ensure good statistics, each simulation was run 100 times, for a total of 400 measurements per data point. The key experimental result, i.e., that longer t_{OFF} rest periods are significantly more effective in reducing α after $t_{\text{ON}} = 1$ ms than $t_{\text{ON}} = 20$ ms charging pulses, is clearly confirmed by calculations (Fig. 2 and Fig. S1). Fig. 3 displays the results of sample simulations. Metallic dendrites grow with random morphologies into equipotential structures held at $V = 0$ Volts, thereby perturbing the uniform electric field prevailing at the beginning of the experiments. The high-curvature dendrite tips act as powerful attractors for the electric field, which by accelerating Li^+ toward their surfaces deplete the electrolyte, further enhancing the field intensity. This strong positive feedback mechanism has its counterpart in the electrolyte regions surrounded by dendrites. The electric field therein nearly vanishes as a consequence of being surrounded by concave equipotential surfaces.³¹ The key feature is that ion displacements from electromigration are proportional to τ_{ON} , whereas diffusive ones increase as $\tau_{\text{ON}}^{1/2}$. Above some critical τ_{ON} value, the depth of the deplete layers will increase to the point at which they could not be replenished during the ensuing rest periods regardless of their duration.

These phenomena are visualized from the computational results shown in Figs. 3-6. Fig. 4 displays the dendrite morphologies created by pulsing at various γ 's. Calculations for longer t_{OFF}

values show marginal improvements, since $\partial[\text{Li}^+]/\partial y$ gradients remain largely unaffected in simulations for $\gamma > 3$. Fig. 5 shows typical morphologies of dendrites consisting of a given number of deposited Li^0 .

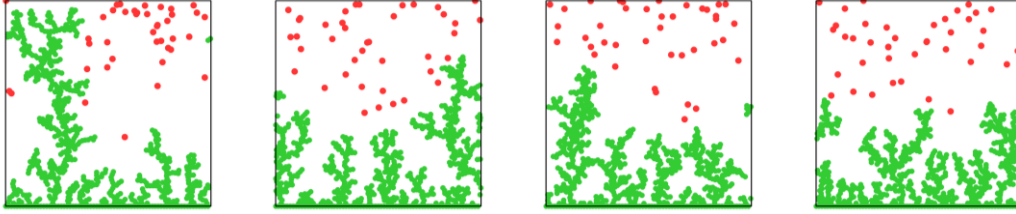


Fig. 3 - Left to right: dendrite morphologies for DC charging, charging with $t_{\text{ON}} = 1$ ms pulses at $\gamma = t_{\text{OFF}}/t_{\text{ON}} = 1, 2$ and 3. Green dots: Li^0 . Red dots: Li^+ .

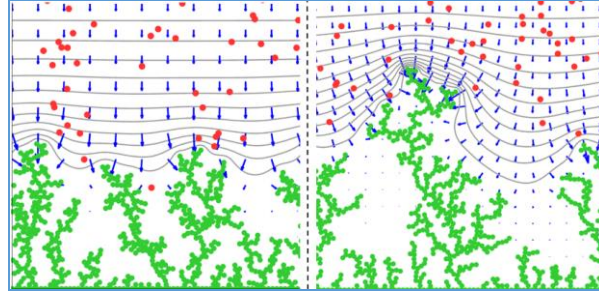


Fig. 4 - Simulations for charging with $t_{\text{ON}} = 1$ ms (left) and $t_{\text{ON}} = 20$ ms (right) at $\gamma = t_{\text{OFF}}/t_{\text{ON}} = 3$. Green dots: Li^0 . Red dots: Li^+ . Gray lines: equipotential contours. Blue vectors: the electric field.

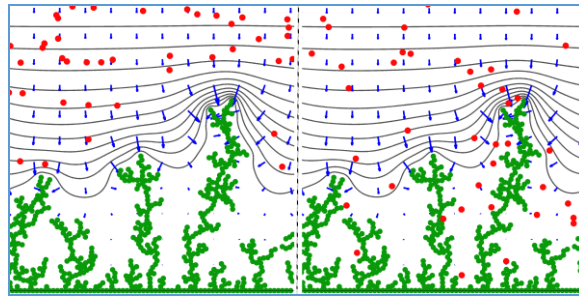


Fig. 5 - Simulations for charging with $t_{\text{ON}} = 1$ ms, $\gamma = 1$ pulses. Left: after a charging pulse. Right: at the end of the successive rest period (right). Green dots: Li^0 . Red dots: Li^+ . Gray lines: equipotential contours. Blue vectors: the electric field.

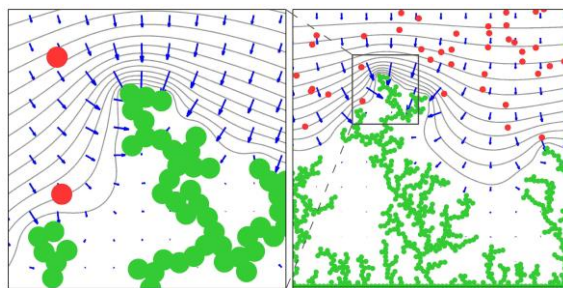


Fig. 6 - Zooming in the tip of the leading dendrite produced by charging with $t_{ON} = 20$ ms, $\gamma = t_{OFF}/t_{ON} = 3$ pulses for 243 ms, i.e., at the end of simulation time. Green dots: Li0. Red dots: Li+. Gray lines: equipotential contours. Blue vectors: the electric field.

Summing up, we have demonstrated that (1) by charging our lithium metal cell with $t_{ON} = 1$ ms, $\gamma = t_{OFF}/t_{ON} = 3$ pulse trains the average dendrite length α is significantly reduced (by ~ 70 %) relative to DC charging, (2) such pulses are nearly optimal for dendrite inhibition because they are commensurate with the relaxation time $\tau_c \sim 3$ ms for the diffusive charging of the electrochemical double layers in our system. Monte Carlo simulations dealing explicitly with lithium ion diffusion, electromigration in time-dependent electric fields and deposition at the cathode were able to reproduce the trends on the experimental effects of the t_{ON} charging and t_{OFF} rest periods on α . Further work along these lines is underway.

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Supporting information – Additional data, experimental and calculational details, is available free of charge via the internet at <http://pubs.acs.org/>

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